

## Radiation Stability of $\text{Ti}_2\text{InC}$ ( $\text{M}_2\text{AX}$ ) Nanolaminates Under He Ions Irradiation – Evaluation Through STEM microscopy

Snejana Bakardjieva<sup>1\*</sup>, Jiri Vacik<sup>2</sup>, Antonino Cannavo<sup>2</sup>, Alena Michalcova<sup>3</sup>, Robert Klie<sup>4</sup>, Xue Rui<sup>4</sup>

<sup>1</sup>. Institute of Inorganic Chemistry of the Czech Academy of Sciences, 205 68 Rez, Czech Republic.

<sup>2</sup>. Nuclear Physics Institute of the Czech Academy of Sciences, 205 68, Czech Republic.

<sup>3</sup>. Department of Metals and Corrosion Engineering, University of Chemistry and Technology, Technicka 5, Prague 6, 166 28, Czech Republic.

<sup>4</sup>. University of Illinois at Chicago, Department of Physics, 845 W Taylor Str., 60 607 Chicago, USA.

\* Corresponding author: snejana@iic.cas.cz

In the 1960s, H. Nowotny and co-workers discovered a family of layered carbides and nitrides, which they called the 'H' phases, now known as the '211' MAX phases (i.e.  $n = 1$ ) [1]. These carbides and nitrides possess unusual combination of chemical, physical, electrical, and mechanical properties, exhibiting both metallic and ceramic characteristics under various conditions which can be well explained by their anisotropic lamellar microstructure [2]. The hexagonal-layered  $\text{Ti}_2\text{InC}$  compound is a member of the lamellar  $\text{M}_2\text{AX}$  family that crystallizes in the hexagonal structure with space group  $\text{P6}_3/\text{mmc}$ .

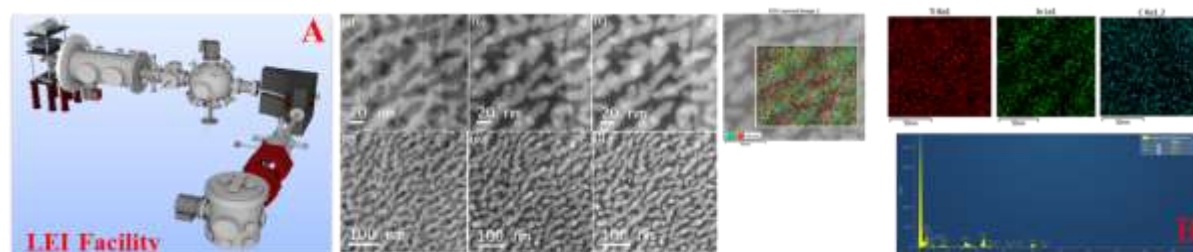
Here, we report for the first time  $\text{Ti}_2\text{InC}$  ( $\text{M}_2\text{AX}$ ) nanolaminates synthesized by an unconventional method utilizing sputtering of individual elements by LEIF (Low Energy Ion Facility) as shown in Fig. 1A. The LEIF facility was utilized without a separation magnet, which means that the MAX phase targets were bombarded with a mixture of  $\text{Ar}^+$  and  $\text{Ar}^{2+}$  ions. The  $\text{Ar}^+$  ions were accelerated to the energy 25 kV, the beam current was kept on a high level of 400  $\mu\text{A}$ . The target holder was designed in a specific way in order (i) to keep the MAX phase elements separated and (ii) to rotate the holder with a variable rotation speed based on a sputtering yield of each MAX phase element. The thickness of the  $\text{Ti}_2\text{InC}$  nanolaminate was found to be about 65 nm by EELS. After sputter deposition, the samples were subsequently annealed in vacuum at 120°C for 24 hrs in order to induce interphase chemical interaction and complete formation of the stoichiometrically-correct  $\text{M}_2\text{AX}$  compounds. The as obtained  $\text{Ti}_2\text{InC}$  samples were further investigated for structural and controlled modification by energy ion beams. In order to analyze the radiation stability in the terms of damage profile, the He ions irradiation was performed. A set of as-prepared samples was irradiated with  $\text{He}^+$  ions with energy 100 keV up to the fluence  $10^{15} \text{ cm}^{-2}$  with the beam current 0.5  $\mu\text{A}$ . Another set of the samples was irradiated with  $\text{He}^+$  ions with energy 100 keV up to the fluence  $10^{17} \text{ cm}^{-2}$  with the beam current 3  $\mu\text{A}$ .

Figure 1 (a-f) shows a series of BF, LAADF and HAADF images of non-irradiated  $\text{Ti}_2\text{InC}$  nanolaminates at low (Fig. 1 a-c) and higher (Fig. 1 d-f) magnifications. As can be seen that the surface of the sample is built by particles organized within the clusters with anisotropic growth. Irradiation with  $\text{He}^+$  ions (100 keV, fluence  $10^{15} \text{ cm}^{-2}$ , beam current 0.5  $\mu\text{A}$ ) does not induce significant changes of roughness (Fig. 2a) and hexagonal structure of  $\text{Ti}_2\text{InC}$  (PDF 54-0505) is preserved (Fig. 2b and FFT analysis as inset). Nevertheless, irradiation defects are clearly observed as compared with the untreated sample: lamellar twinning in  $\text{Ti}_2\text{InC}$  crystals (Fig. 2c) and formation of crystalline  $\text{Ti}_2\text{InC}$  core with concentrated point defects and amorphous shell structure (Fig. 2d). The roughness of the  $\text{Ti}_2\text{InC}$  arises when we used with  $\text{He}^+$  ions with energy 100 keV, fluence  $10^{17} \text{ cm}^{-2}$  and beam current 3  $\mu\text{A}$ . Figure 2e demonstrates

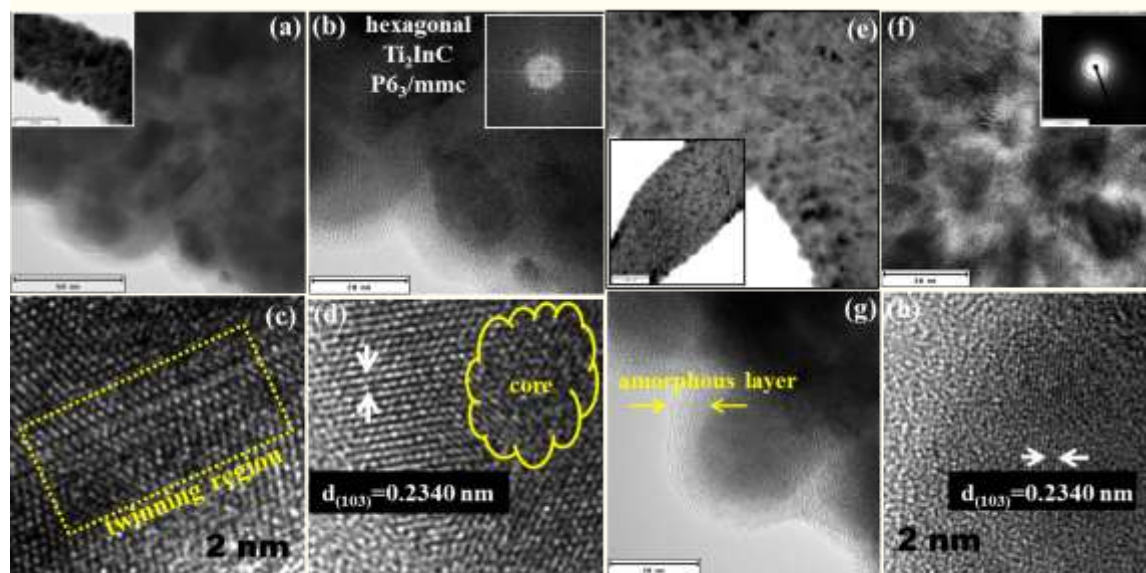
reduced grain size and structural alteration includes deformed nanocrystals and amorphous segments in the  $\text{Ti}_2\text{InC}$  framework. The formation of amorphous thick layer (over 5 nm) have been produced (Fig. 2g); the achievable thickness was increased in comparison with  $\text{He}^+$  ion bombardment with lower fluence. It is important to note that the crystalline nature of  $\text{Ti}_2\text{InC}$  is significantly intact as shown in Fig. 2h. The present results demonstrate suitability of  $\text{Ti}_2\text{InC}$  nanolaminates for radiation and harsh environmental applications.

#### References:

- [1] W Jeitschko, H Nowotny, F Benesovsky, *Journal of the Less Common Metals*. **7** (1964), p.133.  
 [2] M Naguib, *Advanced Materials*. **23** (2011), p. 4248.  
 [3] The authors acknowledge funding from the Ministry of Education, Youth and Sport of the Czech Republic, Project LTAUSA 17128.



**Figure 1.** (A) LEI facility for  $\text{M}_2\text{AX}$  preparation, (a-f) a series of STEM – BF, LAADF and HAADF images showing variation of contrast from the same region at different magnifications (B) EDS analysis and elemental mapping confirming Ti, In and C.



**Figure 2.** Microstructural evolution of  $\text{Ti}_2\text{InC}$  nanolaminate during irradiation by (a-d)  $\text{He}^+$  ions with energy 100 keV up to the fluence  $10^{15} \text{ cm}^{-2}$  and beam current  $0.5 \mu\text{A}$  (e-h)  $\text{He}^+$  ions with energy 100 keV up to the fluence  $10^{17} \text{ cm}^{-2}$  and beam current  $3 \mu\text{A}$ .