

Electron microscopy characterization of W-O multilayers

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In the decorative field, besides the optical characteristics needed for achieving desired colours, the material should have suitable properties able to assure the daily use of the component without significant degradation [1]. Oxides are well known materials with a high wear and corrosion resistance which make them a preferential choice of coatings for environmental protection. However, the intrinsic colours possible to be achieved in a single oxide system are very limited. For example in W-O system, besides the different tonalities of grey, only the dark blue can be obtained for slightly substoichiometric WO_3 compound [2]. The pallet of colours exhibited by WO_3 coatings is due to interferometric effects. Recently, it was shown that playing with the gradient of the chemical composition of W-O layers it was possible to tune the global colour of the coatings, which made this system an attractive candidate for decorative applications [3].

In this study the microstructural characterization of W-O coatings by transmission and scanning electron microscopies (TEM and SEM, respectively) are presented. The coatings were deposited by reactive magnetron sputtering in RGPP mode (reactive gas pulsing process) [4]. SEM micrographs show the multilayer character of the deposited coatings whatever the pulsing parameters used in the deposition were (T – period, and t_{ON} – injection time) [5].

X-ray diffraction revealed that the structure of the coatings deposited by RGPP was in agreement to the multilayer effect, i.e. the detected phases were related to the W-rich and the O-rich layers forming the stacking [5]. All the coatings exhibited a mixture of α -W, β - W_3O and quasi-crystalline W-O (a broad peak centred at $2\theta \approx 25^\circ$, corresponding to the more intense peaks of WO_3) phases attributed to the W layer, a transition layer and the compound layer, respectively [5]. Using TEM, even in high-resolution mode, only 2 layers are possible to be distinguished (Fig. 1a), and the β - W_3O phase could hardly be observed (see Fig. 1b), although electron diffraction (insert of Fig. 1a) confirmed its existence. The W-rich layer showed a fairly crystalline structure in comparison with the featureless aspect of the nanocrystalline O-rich layer.

In order to distinguish and calculate the extension of each of the layers that form the stacking, the cross section of the coatings was analysed by energy filtering images (GIF) which allows evaluating the O concentration across the film thickness. The GIF images were obtained by subtraction of two images taken before and after the O-K edge defined at 530.6 eV, i.e. at -15 eV and +15 eV, respectively. Thus, the new image was generated by the electrons with energy between 515.6 and 545.6 eV as shown in Fig. 2a. By integration of the intensities in a selected area of Fig. 2a, the oxygen distribution across the coating thickness can be accessed (Fig. 2b). In this particular case, the definition of the stacking layers, including the transition layer, can be achieved, allowing to calculate their thickness, even if they are not perfectly distinguished by HRTEM observation. For example, in this case the transition layer is about 45 nm thickness. This information is revealed to be

fundamental for modelling the optical behaviour of the multilayer, since other characterization techniques (e.g. Auger spectroscopy analysis after progressive in-depth etching of the sample) do not allow to establish directly the correspondence between the chemical composition and the depth at which the analysis is being performed [3].

References

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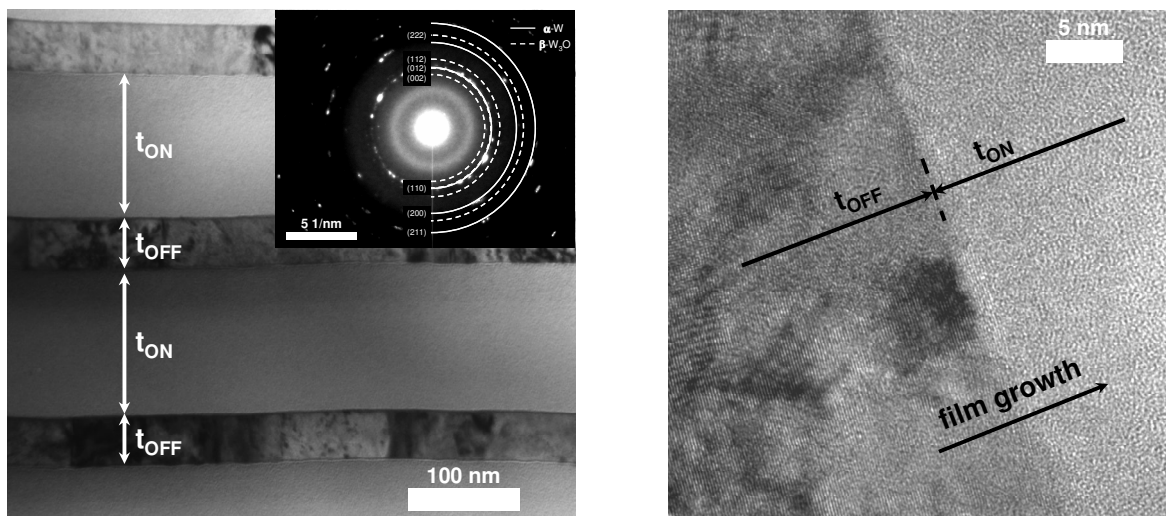


Fig. 1. (a) Cross-section bright field TEM image with selected area electron diffraction pattern in the insert and (b) HR-TEM image of a W-O coating deposited by RGPP ($T=50$ s and $t_{ON} = 25$ s).

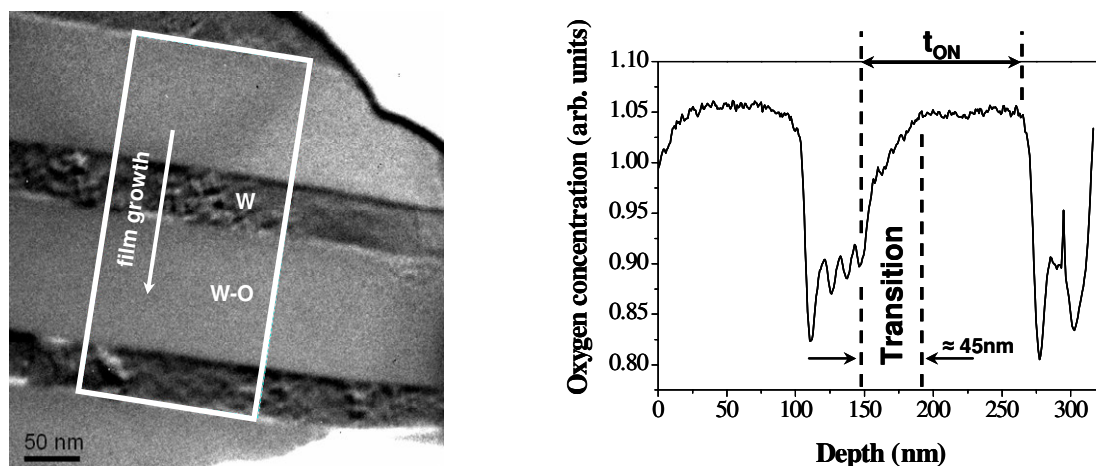


Fig. 2. (a) Oxygen map image obtained from energy filtered TEM of the O-K energy of the coating WO_{50-25} and (b) integration of the intensities from the selected area in (a).