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# **Delaminated and Pit‑rich Nano‑kaolinites obtained via an Intercalation‑etching Method and their Application to Fischer–Tropsch Synthesis**

**Hongxiao Qu · Tong [Liu](http://orcid.org/0000-0002-0233-996X) · Ruijue Hu · Hao Qu · Yichi Zhang · Hui Yang · Haiquan Su · Yue Su**

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**Abstract** Modifed kaolinites possess excellent adsorption properties and, therefore, are regarded widely as potential catalytic components. The use of modifed kaolinites as a catalytic component for Fischer–Tropsch synthesis (FTS) has remained unexplored, however. In the current study, delaminated and pit-rich nano-kaolinite was prepared via acid treatment of N-methylformamide (NMF)-intercalated kaolinite (intercalation-etching strategy), and was used as a support to prepare a cobalt-based FTS catalyst (denoted as 15%-Co-HNKln). Compared with other FTS catalysts, the supports for which were raw kaolinite or directly acid-treated kaolinite, the 15%-Co-HNKln showed several advantages such as large specifc surface area, dispersed Co particles with small particle size, more new active sites, and

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H. Qu · T. Liu · R. Hu · H. Qu · Y. Zhang · H. Yang · H. Su Inner Mongolia Key Laboratory of Chemistry and Physics of Rare Earth Materials, School of Chemistry and Chemical Engineering, Inner Mongolia University, Hohhot 010021, China

### Y. Su  $(\boxtimes)$

School of Ecology and Environment, Inner Mongolia University, Hohhot 010021, China e-mail: suyue@imu.edu.cn

signifcant surface acidity. Given the aforementioned advantages, the 15%-Co-HNKln catalyst demonstrated very good FTS performance. Compared with that of the raw kaolinite-supported FTS catalyst, the CO conversion rate and  $C_5-C_{12}$  hydrocarbon selectivity of 15%-Co-HNKln increased by 20% and 15%, respectively.

**Keywords** Acid treatment · Cobalt-based catalyst · Fischer–Tropsch synthesis · Intercalation · Modifed kaolinite

## **Introduction**

Kaolinite is among the best known clay minerals and an important industrial material. The chemical formula of kaolinite is  $Al_2O_3.2SiO_2.2H_2O$ , and it has a structure of 1:1 dioctahedral layers where each layer consists of a single silica tetrahedral sheet and a single alumina octahedral sheet (Abdullahi & Othman, [2017;](#page-11-0) Gruner, [1932;](#page-11-1) Hendricks, [1936\)](#page-11-2). Much research has been devoted to the modifcation of kaolinite to improve its performance in certain felds (Chen & Lu, [2014;](#page-11-3) Hai & Asuha, [2015\)](#page-11-4). The application of kaolinite is found in diverse felds, and employing modifed kaolinite as a catalytic component is appealing due to its excellent adsorption properties (Li & Dionysiou, [2018,](#page-11-5) Li & Sun, [2020\)](#page-11-6). Catalytic reactions producing renewable fuels are of interest, among which the Fischer–Tropsch synthesis (FTS) is a hotspot owing to its ability to convert syngas (CO and  $H<sub>2</sub>$ ) to valuable chemicals or fuels (Fischer & Tropsch, [1926](#page-11-7)). Exploring the use of kaolinite as a catalytic component for FTS is, thus, a worthwhile venture.

To make kaolinite more suitable for catalytic applications, such as FTS, a proper modifcation is required. Certain characteristics of raw kaolinite, such as small specifc surface area and thick stacking layers, are known to be detrimental to catalytic performance (Qu & Yang, [2023\)](#page-12-0). The most common solution to this problem is to modify the kaolinite via the intercalation-delamination/exfoliation method, which efectively reduces the thickness of stacking layers of kaolinite and leads to increases in the specifc surface area and the number of reactive sites. The intercalation-delamination/exfoliation method has been studied widely since the lattice expansion of kaolin minerals treated with potassium acetate, as investigated by Wada [\(1961](#page-12-1)). One of the intercalation-delamination/ exfoliation agents used commonly is dimethylsulfoxide (DMSO) (Abukhadra & Luqman, [2021](#page-11-8); Tian & Ibrahim, [2020;](#page-12-2) Zuo & Yang, [2018](#page-12-3)). Nevertheless, for certain catalytic reactions such as FTS, the residual sulfur in the DMSO-intercalated kaolinite will poison the catalyst and, therefore, deactivate the catalytic reaction. Introducing non-sulfur intercalation agents would, therefore, be beneficial to FTS. The dielectric properties of intercalated kaolinites were investigated by Orzechowski and Głowinski [\(2006](#page-12-4)) who found that N-methylformamide (NMF) was intercalated almost perfectly, whereas, in the case of potassium acetate, the estimated efficiency was only  $81\%$ . Therefore, for the current study, the non-sulfur NMF is an excellent alternative to the DMSO.

Besides delaminating/exfoliating the kaolinite layers, creating pores/pits on the surface of kaolinite is also favorable to catalytic reactions. The surface pores/pits can not only increase the specifc surface area but also confne the size of the active component (Lim & Aditya, [2018;](#page-12-5) Prieto & De Jongh, [2014](#page-12-6); Wang & Zhang, [2017](#page-12-7)). Hierarchical structure kaolinite nanospheres with abundant mesopores were reported by Zhang  $\&$  Liu ([2019\)](#page-12-8). These kaolinite nanospheres had a much greater specifc surface area than the raw kaolinite and calcined kaolinite, thus showing excellent adsorption performance with high uptake efficiency to Methylene Blue. An  $Au/CeO<sub>2</sub>$ catalyst with a high density of surface pits, which was used for low-temperature CO oxidation was demonstrated by Cai and Guo ([2022\)](#page-11-9). The surface pits confned the Au species and made them resistant to sintering. As a result, stabilized Au species were formed on the CeO<sub>2</sub> support, and the Au/CeO<sub>2</sub> catalyst showed high stability (Cai & Guo, [2022\)](#page-11-9). Hence, delaminated/exfoliated kaolinite with surface pores/ pits are expected to be more suitable for FTS than raw kaolinite.

To obtain a porous/pitted surface, kaolinite is often treated with acid (Gao & Asuha, [2016](#page-11-10); Panda & Singh, [2010\)](#page-12-9); nevertheless the acid-etching process is often inefficient without the delamination/ exfoliation of kaolinite. Two problems need to be solved. First, the efficiency of the acid-etching process needs to be improved via the delamination of kaolinite layers. The frst objective of the present study, therefore, was to develop a new method that integrates the NMF intercalation with the acid-treatment technique (intercalation-etching) by treating the NMF intercalated kaolinite with nitric acid. The hypothesis was that the delamination and etching process could occur simultaneously, which would improve the etching efficiency. The second objective was to load the delaminated and pit-rich nanokaolinite with Co and then investigate its application as a FTS catalyst. The hypothesis was that Co in the surface pits would be beneficial in improving its FTS performance.

## **Materials and Methods**

## Materials

Natural raw kaolinite (Kln) was purchased from Sigma-Aldrich (Burlington, Massachusetts, USA). NMF (analytical reagent, AR) was obtained from Aladdin (Shanghai, China). Nitric acid (AR), cobalt nitrate hexahydrate  $(Co(NO_3)_2.6H_2O, AR)$ , and absolute ethanol were purchased from Tianjin Fengchuan Chemical Reagent Co., Ltd (Tianjin, China). All chemicals were used directly without further purifcation.

Preparation of NMF-Intercalated Kaolinite

10.00 g of kaolinite was dispersed in 100 mL of NMF solution and the mixture was stirred at 80℃ for 24 h. The resulting product was collected by centrifuging

and washed several times with ethanol. Then the solid was dried at 60°C for 12 h to obtain a NMF-kaolinite intercalation complex, which is denoted as NMF-Kln.

## Acid Treatment of Kaolinite Samples

5.00 g raw kaolinite was added to a 100 mL fask, and then 50 mL of 2.5 mol/L nitric acid was added. The mixture was stirred magnetically at 50℃ for 24 h. Then the resulting product was centrifuged and dried at 60°C for 24 h. The sample obtained was labeled as  $HNO<sub>3</sub>$ -Kln. To make a comparison, the NMF-Kln was also treated with nitric acid, which shared the same procedure with the acid treatment of  $HNO_3$ -Kln, and the product was denoted as  $HNO<sub>3</sub>-NMF-Kln$ .

## Preparation of FTS Catalysts

The FTS catalysts were prepared by loading Co particles on the kaolinite samples through the wet impregnation method. Typically, 0.89 g of  $Co(NO_3)_2.6H_2O$ was dissolved in 10.00 mL of deionized water to prepare a salt solution. Subsequently, 1.00 g of Kln was added to the salt solution and stirred at room temperature for 1 h. Then, the mixture was treated with ultrasonic-assisted impregnation for 1 h. The sample obtained was retrieved by centrifugation and dried at 110℃ for 24 h. Subsequently, the sample was crushed to powder and heated from room temperature to 350<sup>°</sup>C with the heating rate of  $2^{\circ}$ C min<sup>-1</sup>. The calcination process was maintained for 4 h at  $350^{\circ}$ C in air. As a result, the Kln was loaded with  $Co<sub>3</sub>O<sub>4</sub>$  particles, which was the precursor of a FTS catalyst. Because the nominal loading content of Co particles was 15 wt.%, the precursor was named as  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln. The  $HNO_3-Kln$  and  $HNO_3-NMF-Kln$  were treated with the same procedure and the precursor FTS catalysts achieved were denoted as  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln and  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln, respectively.

To convert the precursors into freshly reduced FTS catalysts, the  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln,  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln, and  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln went through an in situ reduction at 380°C for 10 h under a  $H_2$  atmosphere. Because the  $Co<sub>3</sub>O<sub>4</sub>$  on the precursors was reduced to  $Co(0)$  after pretreating with  $H<sub>2</sub>$ , the freshly reduced FTS catalysts were labeled as 15%-Co-Kln, 15%-Co-HKln, and 15%-Co-HNKln, respectively.

## Characterization

X-ray difraction patterns (XRD) were obtained on a Malvern Panalytical (Almelo, Netherlands) X'Pert PRO type diffractometer equipped with target  $CuKa$ irradiation ( $\lambda$ =0.15418 nm) at 40 kV and 40 mA, with a scanning speed and slit width of 0.4°2θ/s and 55 mm, respectively, and without the use of a monochromator. The samples characterized by XRD analysis were crushed into powder form. Fouriertransform infrared (FTIR) spectra were obtained using a Thermo Fisher Scientifc (Waltham, Massachusetts, USA) Nicolet 6700 FTIR Spectrometer with a DTGS/KBr detector; the resolution and number of scans were  $2 \text{ cm}^{-1}$  and  $40$ , respectively. The samples characterized by FTIR analysis were diluted about 200 times with KBr, and then pressed into tablets under 2 MPa of pressure. The morphology of the samples was examined using a Hitachi (Tokyo, Japan) S-4800 scanning electron microscope (SEM) and a FEI Tecnai  $G^2$  F20 field-emission transmission electron microscope (TEM). The samples characterized by SEM and TEM analysis were dispersed in 2 mL of ethanol (1 mg) and treated ultrasonically for 30 min. The elemental analysis of catalyst surface was carried out by using a Bruker (Billerica, Massachusetts, USA) energy-dispersive X-ray spectroscopy (EDS) equipped on the HITACHI S-4800 scanning electron microscope. The chemical composition was analyzed by X-ray fuorescence (XRF, ZSX model) manufactured by Rigaku (Tokyo, Japan). The samples characterized by XRF were diluted about 20 times with boric acid and then pressed into tablets. The thermal mass loss of the sample  $(0.05 \text{ g})$  was analyzed on a Hitachi (Tokyo, Japan) EXSTAR TG/ DTA 7300 Thermogravimetry–Diferential Thermal Analyzer under a  $N_2$  flow (15 mL/min) with a heating rate of 5°C/min. The amount of elemental Co in the catalysts was determined by the Thermo Jarrell Ash IRIS Advantage inductively coupled plasma-optical emission spectrometer (ICP-OES) manufactured by the Thermo Fisher Scientifc (Waltham, Massachusetts, USA). The samples analyzed by ICP-OES were dispersed into aqua regia before testing.  $H_2$ -TPR (temperature programmed reaction) and CO-TPD (temperature-programmed desorption) were obtained using a ChemAutoII 2920 automated chemisorption analyzer from Micromeritics Instrument Corporation (Norcross, Georgia, USA). The samples characterized by the  $H_2$ -TPR and CO-TPD analysis were crushed into powder form and purged at 200°C under an Ar atmosphere for 2 h.

The average thickness of kaolinite samples and the particle size of  $Co<sub>3</sub>O<sub>4</sub>$  on the precursors of FTS catalysts were calculated by the Scherrer equation (Scherrer, [1918](#page-12-10)):

$$
L = K \cdot \lambda / \beta \cdot \cos \theta
$$

where *L* represents the crystallite size, *K* is a dimensionless shape factor,  $\lambda$  represents the X-ray wavelength, β represents the full width at half maximum of peaks, and θ is the Bragg angle.

The intercalation ratio of the NMF-kaolinite intercalation complex (NMF-Kln) was calculated by the intensity of XRD refections (Wiewióra, [1969\)](#page-12-11):

$$
CO conversion (\%) = \frac{M_{\text{CO(inlet)}} - M_{\text{CO(off-gas)}}}{M_{\text{CO(inlet)}}} \times 100\%
$$

The product selectivity of the freshly reduced FTS catalysts was calculated using the following equation:

$$
S_i\ (\%) = \frac{C_i}{\sum C_i} \times 100\%
$$

where  $M_{\text{CO (inlet)}}$  and  $M_{\text{CO (off-gas)}}$  represent moles of CO at the inlet and outlet, respectively.  $C_i$  represents the moles of C of the product *i*.

Intercalation Ratio =  $[I_{(001) \text{complex}}/(I_{(001) \text{complex}} + I_{(001) \text{kalinite}}] \times 100\%$ 

where  $I_{(001)complex}$  represents the basal peak intensity of NMF-kaolinite intercalation complex, and *I*(001)kaolinite represents the basal peak intensity of unexpanded kaolinite component in the complex.

## Catalytic Evaluation

0.8 g of each freshly reduced FTS catalyst was used for the evaluation. The catalytic performance was evaluated under reaction conditions of  $H<sub>2</sub>/CO=2:1$ ,  $T=220^{\circ}$ C, and  $P=2.0$  MPa. The composition of offgas was detected by an online gas chromatography (GC) system (Shimadzu GC-2014C, Japan).

### **Results and Discussion**

Characterization of Kaolinite Samples

The powder XRD was conducted to identify the crystalline structure of the Kln,  $HNO<sub>3</sub>-Kln$ , NMF-Kln, and  $HNO<sub>3</sub>-NMF-Kln$  (Fig. [1\)](#page-3-0). In the case of Kln, the refection at 12.39°2θ corresponded to the (001) plane of kaolinites, which revealed the stacking degree of layers in the *c*-axis direction (Zuo and Yang, [2018](#page-12-3)). The narrow reflection of the (001) plane of Kln indicated a high stacking degree of kaolinite layers. As for  $HNO<sub>3</sub>$ -Kln, nearly all of the reflections from Kln were retained after the acid-treatment process, and



<span id="page-3-0"></span>**Fig. 1** XRD patterns of **a** kaolinite samples treated by different approaches and **b**  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln,  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln, and  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln catalyst precursors

moreover, the narrow (001) reflection of  $HNO<sub>3</sub>$ -Kln indicated that the stacking degree of layers was hardly afected by the direct acid-treating process. After the intercalation of NMF, the refection of the (001) plane of NMF-Kln shifted to a lower angle (8.23°2θ), which indicated an expansion of the interlayer space, implying that the NMF had been intercalated successfully into the interlayer space of kaolinites (Sugahara & Kato, [1990](#page-12-12)). According to the formula described in the materials and methods section, the intercalation ratio of NMF was 98%. After treating NMF-Kln with  $HNO<sub>3</sub>$  (sample  $HNO<sub>3</sub>-NMF-KIn$ ), the (001) reflection shifted to a higher angle of 12.39°2θ, which was due to the elimination of NMF and the subsequent shrinkage of interlayer space. The FWHM of the (001) reflection of  $HNO<sub>3</sub>-NMF-Kln$  was obviously larger than that of Kln (0.53 vs 0.18); this was due to the delamination of kaolinites caused by the breakup of hydrogen bonds between the kaolinite layers. According to the Scherrer equation, the average thickness of  $HNO<sub>3</sub>-NMF-Kln$  was calculated as 14 nm. In addition, all of the kaolinite samples exhibited the refection located at 62.26°2θ, which was attributed to the (060) plane of kaolinite, refecting the character of a dioctahedral clay mineral (Letaief & Detellier, [2011\)](#page-11-11). Clearly, the typical dioctahedral structure was preserved in both direct acid-treatment and intercalation-etching process.



<span id="page-4-0"></span>**Fig. 2 a,d** SEM and **g** TEM images of Kln; **b**,**e** SEM, and **h** TEM images of HNO3-Kln; **c**,**f**, and **i** TEM images of HNO3-NMF-Kln

 $V$  and  $V$  and  $V$  $\bigcircled{2}$  Springer

The morphology of the kaolinite samples was characterized by SEM and TEM (Fig. [2\)](#page-4-0). The sample Kln (Fig. [2a](#page-4-0), d) presented a well-ordered pseudohexagonal lamellar structure with thick stacking layers. After the direct acid-treatment process, the  $HNO<sub>3</sub>$ -Kln exhibited negligible delamination (Fig. [2](#page-4-0)b) and  $2e$  $2e$ ). The stacking layers of  $HNO<sub>3</sub>-KIn$  were still thick, with relatively small lamellae due to the etching process. After the intercalation of NMF, the morphology of the sample NMF-Kln exhibited few changes compared with that of Kln (as shown in Supplementary Information Fig. S1). For  $HNO<sub>3</sub>-NMF-KIn$ , an obvious delamination was observed. As shown in Figs S2 and 2f, the thickness of stacking layers of  $HNO<sub>3</sub>-NMF-KIn$  was reduced significantly compared with that of Kln, which was consistent with the XRD result. Meanwhile, Fig. [2c](#page-4-0) shows that the edges of kaolinite nanolayers in  $HNO<sub>3</sub>-NMF-Kln$  were partially curled and slightly damaged, which was due to the break-up of hydrogen bonds and leaching of the surface during the acid-treatment process. Importantly, the TEM images demonstrated that the surface of  $HNO<sub>3</sub>$ -Kln (Fig. [2](#page-4-0)h) and  $HNO<sub>3</sub>$ -NMF-Kln (Fig. 2i) had many small pits (light spots in the fgures) while Kln did not (Fig. [2g](#page-4-0)). The diameter of these surface pits ranged from~3 to 17 nm for both samples. The formation of surface pits was due to the removal of Al from Al-O(OH) in kaolinites by  $H^+$  during acid treatment (Hu & Yang, [2013](#page-11-12)).

To determine the functional groups in the kaolinite samples, FTIR spectra were collected (Fig. [3\)](#page-6-0). In the raw kaolinites, the band at  $3620 \text{ cm}^{-1}$  belonged to the OH-stretching modes of inner hydroxyl groups, while the bands at 3695 and 3669  $cm^{-1}$  were attributed to inner-surface hydroxyl groups (Zhu & Yan, [2016\)](#page-12-13). Meanwhile, the bands at 1101, 1033, and  $1009 \text{ cm}^{-1}$  were assigned to the stretching vibration mode of Si-O bonds. After  $HNO<sub>3</sub>$  treatment, the FTIR spectrum of  $HNO<sub>3</sub>-Kln$  was similar to that of Kln, with no new bands and with all original bands still present. Nevertheless, the relative intensity of the bands in  $3620-3695$  cm<sup>-1</sup> was weakened for  $HNO<sub>3</sub>-KIn$ , implying that both types of the hydroxyl groups were reduced due to the removal of Al from Al–O(OH). After NMF intercalation, the band at 3695  $\text{cm}^{-1}$  in Kln shifted to 3678  $\text{cm}^{-1}$ in NMF-Kln, which was due to the formation of new hydrogen bonds during NMF intercalation (Komori & Kuroda, [1998](#page-11-13)), proving that NMF was intercalated successfully into the interlayers of kaolinites. Meanwhile, in NMF-Kln, the bands at 2908 and  $1350 \text{ cm}^{-1}$  belonged to the symmetric stretching and bending vibrations of  $CH<sub>3</sub>$ , respectively (Tunney & Detellier, [1997](#page-12-14)), the bands at 3419 and  $1529$  cm<sup>-1</sup> were attributed to the stretching and bending vibrations of N–H, respectively, while the band at  $1682 \text{ cm}^{-1}$  was assigned to the stretching vibration of  $C=O$  (Kelleher & O'Dwyer, [2002a](#page-11-14), [2002b](#page-11-15)). All of the bands mentioned above proved further the successful intercalation of NMF into the interlayers of kaolinites. As for  $HNO<sub>3</sub>-NMF-Kln$ , a weak NMF band at  $1684 \text{ cm}^{-1}$  was noted in the FTIR spectrum, and the band associated with the new hydrogen bond  $(3678 \text{ cm}^{-1})$  in NMF-Kln shifted to 3695 cm<sup>-1</sup> in  $HNO<sub>3</sub>-NMF-Kln$ , which indicated the breaking of the new hydrogen bond formed during NMF intercalation. The FTIR data of  $HNO<sub>3</sub>-NMF-Kln$  above demonstrated that most of the NMF was eliminated after  $HNO<sub>3</sub>$  treatment. More importantly, the peak area ratio of Si-O bonds  $(1009 \text{ to } 1101 \text{ cm}^{-1})$  to inner and inner-surface hydroxyl groups (3620 to 3695  $\text{cm}^{-1}$ ) of HNO<sub>3</sub>-NMF-Kln was greater than that of  $HNO_3$ -Kln (4.5 VS 4.2), which indicated that more  $Al_2O_3$  was leached for  $HNO_3$ -NMF-Kln due to a more efficient etching process.

The three kaolinite samples were analyzed by XRF (Table [1\)](#page-6-1). The concentration of metal elements is given in the form of oxides. Clearly, the raw kaolinite (Kln) was composed mainly of  $SiO_2$  and  $Al_2O_3$ , with small amounts of other metal oxides presented as impurities. After direct acid treatment of raw kaolinite, the chemical composition of  $HNO_3-Klh$  was changed slightly, which was due to an insufficient etching process. By contrast, the  $HNO<sub>3</sub>-NMF-Kln$ showed obvious change in its chemical composition, with  $SiO<sub>2</sub>$  increased to 70.10% while  $Al<sub>2</sub>O<sub>3</sub>$  decreased to 26.14%, which indicated that  $Al_2O_3$  was leached efficiently and selectively during the etching of NMF-Kln. Evidently, the expanded interlayer space of NMF-Kln enabled a more efficient leaching of the internal aluminum atoms due to more nitric acid being able to enter the interlayer space of kaolinite, leading to the sharp decrease in the amount of  $Al_2O_3$ in  $HNO<sub>3</sub>-NMF-Kln$ . The XRF result was consistent with the TEM and FTIR analysis.

The thermal mass loss of the kaolinite samples was presented as TG and DTG curves (Fig. [4\)](#page-7-0). As can be seen, the Kln,  $HNO_3-KIn$ , and  $HNO_3-NMF-KIn$ 



<span id="page-6-0"></span>**Fig. 3** FTIR spectra of various kaolinite samples

exhibited a mass loss at 500, 494, and 487℃, respectively, all of which were attributed to the dehydroxylation of kaolinite (Horváth & Cseh, [2003](#page-11-16)). Note that the  $HNO<sub>3</sub>-NMF-KIn$  not only showed the lowest dehydroxylation temperature among the three samples, but also exhibited a continuous mass loss from 50 to 600℃. The reason for this phenomenon was that more hydroxyl groups in kaolinite were exposed on the surface for  $HNO_3-NMF-KIn$  due to the delamination of the layers, thus less energy was required for the dehydroxylation process because these exposed hydroxyl groups could not form hydrogen bonds between the adjacent layers. As for NMF-Kln, two mass-loss peaks could be observed from the TG and DTG curves. The frst mass loss at 135°C was attributed to the decomposition of NMF between

<span id="page-6-1"></span>**Table 1** Chemical composition (wt.%) of the kaolinite samples analyzed by XRF

Samples Chemical Composition	K <sub>ln</sub>	$HNO3$ -Kln	$HNO3-NMF-KIn$
SiO <sub>2</sub>	57.11	60.85	70.10
$\text{Al}_2\text{O}_3$	38.80	35.52	26.14
$Fe_2O_3$	0.58	0.60	0.27
TiO <sub>2</sub>	0.59	0.80	0.84
Na <sub>2</sub> O	0.63	0.13	0.02
$K_2O$	2.16	2.02	2.54
CaO	0.13	0.08	0.09

the kaolinite layers (Adams, [1978](#page-11-17); Li & Komarneni, [2017\)](#page-11-18), while the second mass loss, which appeared at 504°C, was assigned to the dehydroxylation of kaolinites, and this temperature was greater than for the other samples. The reason for this increased dehydroxylation temperature was the breaking of new hydrogen bonds formed during NMF intercalation, where more energy was required (Kelleher & O'Dwyer, [2002b;](#page-11-15) Li & Komarneni, [2017](#page-11-18)). The results above demonstrated further the successful intercalation and removal of NMF, as well as the delamination of kaolinite layers.

To investigate the specifc surface area and textural structure of the kaolinite samples, BET analysis was conducted. As the  $N_2$ -adsorption/desorption isotherms show (Fig.  $5$ ), all of the kaolinite samples exhibited a type II isotherm, which represents a nonporous or macroporous solid. In the present cases, all of the samples were macroporous, and the majority of the pores were generated by the accumulation of delaminated kaolinite layers. In addition,  $HNO<sub>3</sub>-NMF-KIn$  had the largest adsorption volume among all the kaolinite samples, while Kln had the lowest. Interestingly, the absorption volume of NMF-Kln was close to that of  $HNO_3-NMF-Kln$ , which was probably due to the expansion of interlayer space caused by the NMF intercalation.

The specifc surface area and pore volume of Kln,  $HNO<sub>3</sub>-Kln$ , NMF-Kln, and  $HNO<sub>3</sub>-NMF-Kln$  are listed in Table [2.](#page-7-2) Compared with Kln, the  $HNO<sub>3</sub>$ -Kln,  $NMF-Kln$ , and  $HNO<sub>3</sub>-NMF-Kln$  exhibited increased specifc surface area and pore volume. Note that the  $HNO<sub>3</sub>-NMF-KIn possessed the largest specific surface$ area and pore volume. A signifcant increase in specific surface area for  $HNO<sub>3</sub>-NMF-Kln$  was due to the delamination of kaolinite layers and generation of surface pits, and the delamination process was critical. The increased pore volume for  $HNO<sub>2</sub>-NMF-Kln$  was due to the generation of more accumulation pores formed by the accumulation of delaminated kaolinite layers. According to the BET analysis, the  $HNO<sub>3</sub>-NMF-Kln$ would be expected to be the most efficient catalyst support among all the kaolinite samples.

## Characterization of the Precursors of FTS Catalysts

After Kln,  $HNO_3-KIn$ , and  $HNO_3-NMF-KIn$  were loaded with cobalt nitrate hexahydrate and then calcined, the precursors of FTS catalysts were obtained,



<span id="page-7-0"></span>**Fig. 4 a** TG and **b** DTG curves of Kln, HNO<sub>3</sub>-Kln, NMF-Kln, and HNO<sub>3</sub>-NMF-Kln

which were denoted as  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln,  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln, and  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln, respectively. Powder XRD was conducted to analyze the crystalline structure of the three precursors (Fig. [1b](#page-3-0)). Obviously, all of the refections of raw kaolinites were preserved in the precursors, which indicated that the structure of kaolinite was not destroyed. Meanwhile, new refections at 18.99°2θ (111), 38.54°2θ (222), and 65.23°2θ (440) appeared in all precursors, which were assigned to  $Co_3O_4$  (PDF: 01–078-1969). According to the Scherrer equation, the average size of  $Co<sub>3</sub>O<sub>4</sub>$  particles of  $15\% - \text{Co}_3\text{O}_4$ -HKln and  $15\% - \text{Co}_3\text{O}_4$ -HNKln was~24 and 23 nm, respectively, which was obviously smaller than that of  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln (33 nm). The reason for the reduced  $Co<sub>3</sub>O<sub>4</sub>$  particle size was probably due to the surface pits of  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln and  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln. When Co<sub>3</sub>O<sub>4</sub> particles were



<span id="page-7-1"></span>**Fig. 5**  $N_2$ -adsorption/desorption isotherms of Kln,  $HNO_3-Klh$ , NMF-Kln, and  $HNO<sub>3</sub>$ -NMF-Kln

embedded in these pits, the growth of particles was restricted.

The morphology and structure of the  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln,  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln, and  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln precursors were demonstrated in the SEM, TEM, and HRTEM images (Fig. [6](#page-8-0)). As shown in Fig. [6a](#page-8-0)–c, the surfaces of the three precursors were covered with small particles which were assigned to  $Co_3O_4$  due to the interplanar distances of these small particles being 0.286 and 0.245 nm (as shown in Fig.  $6f$ , corresponding to the  $(022)$  and  $(311)$ lattice planes of  $Co<sub>3</sub>O<sub>4</sub>$ , respectively. Besides, the  $Co<sub>3</sub>O<sub>4</sub>$  particles showed obvious aggregation for  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln, while for  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln and  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln the Co<sub>3</sub>O<sub>4</sub> particles were relatively dispersed, especially for  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln. The reason  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln demonstrated the greatest dispersion of  $Co<sub>3</sub>O<sub>4</sub>$  particles was that it had the largest specifc surface area. The TEM images demonstrated that the  $Co<sub>3</sub>O<sub>4</sub>$  particles of  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln (Fig. [6d](#page-8-0)) and  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln (Fig. [6](#page-8-0)e) were quite uneven in size. The signifcantly smaller  $Co<sub>3</sub>O<sub>4</sub>$  particles were probably embedded in the surface pits, while the obviously larger ones

<span id="page-7-2"></span>Table 2 Specific surface area and pore volume of Kln,  $HNO<sub>3</sub>$ -Kln, and  $HNO<sub>3</sub>$ -NMF-Kln

	Specific surface area Pore volume $\text{m}^2\text{/g}$	$(m^3/g)$	
K <sub>ln</sub>	9.75	0.04	
$HNO3$ -Kln	38.43	0.07	
NMF-Kln	41.25	0.08	
$HNO3$ -NMF-Kln	47.53	0.10	



**Fig.** 6 SEM images (at various magnifications) of **a**  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln, **b**  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln, and **c**  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln; TEM images of **d** 15%-Co<sub>3</sub>O<sub>4</sub>-HKln, **e** 15%-Co<sub>3</sub>O<sub>4</sub>-HNKln, and **f** HRTEM image of 15%-Co<sub>3</sub>O<sub>4</sub>-HNKln

<span id="page-8-0"></span>were simply dispersed on the surface. The embedding strategy has an advantage in controlling the size of metal particles and preventing the migration of metal, thus the small  $Co<sub>3</sub>O<sub>4</sub>$  particles embedded in the surface pits are very likely to improve the FTS performance. Based on the discussion above, the  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln is expected to be an inferior precursor to  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln and  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln. Furthermore, due to the delaminated kaolinite layers, the thickness of  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln was less than that of  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln (as shown in Fig. S3), which was consistent with the thickness of Kln and  $HNO<sub>3</sub>-NMF-Kln support.$ 

EDS mapping was conducted for the three precursors (Fig. S4). The actual distributions of elemental Co revealed that the  $Co<sub>3</sub>O<sub>4</sub>$  particles of 15%-Co<sub>3</sub>O<sub>4</sub>-Kln showed obvious aggregation, while  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln had the most dispersed particles. This result was consistent with the SEM analysis. Meanwhile, as shown in the EDS analysis, the three precursors exhibited similar peak values of Co, which demonstrated similar  $Co<sub>3</sub>O<sub>4</sub>$  loading amounts among the three precursors.

To determine further the amount of Co in the three precursors of FTS catalysts, ICP-OES was conducted. The loading amounts of Co of  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln,  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln, and  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln were 16.5, 15.9, and 15.7 wt.%, respectively (Table [3](#page-8-1)). The similar loading amounts of the three precursors indicated that the loading process was scarcely afected by either direct acid treatment or by the intercalationetching strategy.

 $H_2$ -TPR profiles for all precursors are shown in Fig. [7a](#page-9-0). For sample  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-Kln, two peaks centered at 280 and 315°C were observed, which corresponded to the reduction of  $Co^{3+}$  to  $Co^{2+}$  and  $Co^{2+}$ to  $Co^0$ , respectively (Irandoust & Haghtalab, [2017](#page-11-19)).

<span id="page-8-1"></span>**Table 3** Loading of Co in the catalyst precursors analyzed by ICP-OES

Catalysts	$15\% - \text{Co}_3\text{O}_4$ -	$15\% - \text{Co}_3\text{O}_4$ -	$15\% - \text{Co}_3\text{O}_4$ -
	Kln	HKln	<b>HNK</b> In
Co loading $(wt,\%)$	16.5	15.9	15.7



<span id="page-9-0"></span>**Fig. 7** a H<sub>2</sub>-TPR of the 15%-Co<sub>3</sub>O<sub>4</sub>-Kln, 15%-Co<sub>3</sub>O<sub>4</sub>-HKln, and 15%-Co<sub>3</sub>O<sub>4</sub>-HNKln precursors and **b** CO-TPD spectra of 15%-Co-Kln, 15%-Co-HKln, and 15%-Co-HNKln catalysts

As for  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HKln, the reduction temperatures of  $Co^{3+}$  to  $Co^{2+}$  and  $Co^{2+}$  to  $Co^{0}$  appeared at 270 and 307°C, respectively. A small peak appeared at the right shoulder of the 307°C peak, which was assigned to the reduction of the small CoO particles embedded in the surface pits. As is well known, smaller particles have greater interactions with the support than larger ones, thus a higher temperature was required to reduce the CoO particles (Akbar-zadeh & Johan, [2018\)](#page-11-20). For  $15\%$ -Co<sub>3</sub>O<sub>4</sub>-HNKln, the  $Co^{2+}$  to  $Co^0$  reduction peak shifted to a much higher temperature of 465℃, which indicated that the interaction between CoO particles and support was much stronger in this sample. In other words, the size of CoO was much smaller for  $15\% - \text{Co}_3\text{O}_4$ -HNKln because this sample had a larger specifc surface area and more surface pits, which was benefcial to the dispersion and confnement of CoO particles.

After the three precursors were converted to freshly reduced FTS catalysts (15%-Co-Kln, 15%-Co-HKln, and 15%-Co-HNKln), CO-TPD was conducted to characterize the adsorption and desorption of CO on the surface of the catalysts (Fig. [7b](#page-9-0)). Through this characterization, the active sites on the surface of the FTS catalysts can be investigated. As shown in the CO-TPD pattern, only one CO desorption peak centered at 522℃ was observed in 15%-Co-Kln, while two major peaks were observed for both 15%-Co-HKln and 15%-Co-HNKln. The desorption peak which appeared at  $\sim$  520°C in all of the samples belonged to the active sites of normally dispersed Co particles, while the new desorption peak which appeared at ~730℃ in 15%-Co-HKln and 15%-Co-HNKln was assigned to the new active sites of highly dispersed Co particles with small particle size embedded in surface pits. Note that the peak area associated with the new active sites was much larger for 15%-Co-HNKln, which indicated that this sample possessed the largest number of new active sites (or surface pits), implying that 15%-Co-HNKln went through the most efficient acid-etching process. Moreover, rich surface pits were benefcial to the generation of small particles and thus it was reasonable that 15%-Co-HNKln exhibited the greatest CoO reduction temperature as shown in the TPR.

### Catalytic Performance

The catalytic performance of freshly reduced FTS catalysts was evaluated under the reaction conditions



<span id="page-10-0"></span>**Fig. 8** CO conversion rate and product selectivity of 15%-Co-Kln, 15%-Co-HKln, and 15%-Co-HNKln catalysts

of H<sub>2</sub>/CO=2:1,  $T = 220^{\circ}$ C, and  $P = 2.0$  MPa (Fig. [8,](#page-10-0) Table [4\)](#page-10-1). Clearly, the CO conversion rate and  $C_5-C_{12}$  hydrocarbon selectivity of 15%-Co-HKln and 15%-Co-HNKln were signifcantly greater than those of 15%-Co-Kln, and 15%-Co-HNKln demonstrated the best FTS performance. The reason that 15%-Co-HNKln exhibited the highest CO conversion rate was that it had the largest specifc surface area, the most dispersed Co particles with smallest particle size, as well as the richest new active sites. Besides, as concluded in the CO-TPD analysis above, the 15%-Co-HNKln went through the most efficient acid-etching process and, thus, the surface acidity of this sample was the greatest in the samples. Because signifcant surface acidity was benefcial to the generation of hydrocarbons with longer chains (Wang and Hu, [2019](#page-12-15)), the 15%-Co-HNKln showed the greatest  $C_5-C_{12}$  hydrocarbon selectivity.

## **Conclusions**

In the present study, delaminated and pit-rich nanokaolinite was obtained via the intercalation-etching strategy, and FTS catalyst was prepared after loading Co particles on the nano-kaolinites (15%-Co-HNKln). Compared with other FTS catalysts, the supports for which were raw kaolinite or directly acid-treated kaolinite, the 15%-Co-HNKln catalyst showed several advantages such as the largest specifc surface area, the most dispersed Co particles with smallest particle size, the richest new active sites, and the greatest surface acidity. Because of the advantages mentioned above, the 15%-Co-HNKln showed the best FTS performance among the three catalysts, with the highest CO conversion rate of 53.47% and the highest  $C_5-C_{12}$  hydrocarbons (gasoline) selectivity of 51.49%. Compared with that of raw

Catalyst	CO	CH <sub>A</sub>	$C_2-C_4$	$C_5 - C_{12}$	$C_{13}$ - $C_{19}$	Oxides
	Conv. <sup>a</sup> $(\%)$	Selec. <sup>b</sup> $(\%)$	Selec. $(\%)$	Selec. $(\%)$	Selec. $(\%)$	Selec. $(\%)$
15%-Co-Kln	32.9	28.3	25.4	37.4	1.7	7.2
$15\%$ -Co-HKln	43.2	25.9	17.9	45.8	5.0	5.4
15%-Co-HNKln	53.5	22.3	17.0	51.5	4.8	4.4

<span id="page-10-1"></span>**Table 4** Catalytic performance of 15%-Co-Kln,15%-Co-HKln, and 15%-Co-HNKln catalysts

a"Conv." represents "Conversion"

<sup>b</sup>"Selec." represents "Selectivity"

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kaolinite-supported FTS catalyst, the CO conversion rate and  $C_5-C_{12}$  hydrocarbons selectivity of 15%-Co-HNKln showed increases of 20 and 15%, respectively. Clearly, the kaolinite could be modified efficiently by the intercalation-etching strategy and, therefore, the FTS performance of kaolinite-supported catalyst could be improved signifcantly. This study provides new guidance to the modifcation and catalytic application of kaolinite.

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**Authors' Contributions** Hongxiao Qu: Writing—original draft. Tong Liu: Formal analysis. Ruijue Hu: Writing—review and editing. Hao Qu: Formal analysis. Yichi Zhang: Formal analysis. Hui Yang: Formal analysis. Haiquan Su: Funding acquisition, Data curation, Writing—review and editing. Yue Su: Conceptualization, Formal analysis, Writing—review and editing.

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**Data Availability** All data generated or analysed during this study are included in this published article.

**Code Availability** Not applicable.

**Declarations**

**Competing Interests** On behalf of all authors, the corresponding author states that there are no conficts of interest.

## **References**

- <span id="page-11-0"></span>Abdullahi, T., & Othman, M. H. D. (2017). A review on sustainable synthesis of zeolite from kaolinite resources via hydrothermal process. *Advanced Powder Technology, 28*(8), 1827–1840.
- <span id="page-11-8"></span>Abukhadra, M. R., & Luqman, M. (2021). Enhancing the removal of organic and inorganic selenium ions using an exfoliated kaolinite/cellulose fbres nanocomposite. *Carbohydrate Polymers, 252*, 117163.
- <span id="page-11-17"></span>Adams, J. (1978). Diferential scanning calorimetric study of the kaolinite: N-methylformamide intercalate. *Clays and Clay Minerals, 26*, 169–172.
- <span id="page-11-20"></span>Akbarzadeh, O., & Johan, M. R. (2018). Efects of cobalt loading, particle size, and calcination condition on Co/CNT

catalyst performance in Fischer-Tropsch reactions. *Symmetry, 11*(1), 7.

- <span id="page-11-9"></span>Cai, Y. F., & Guo, Y. (2022). Surface pits stabilized Au catalyst for low-temperature CO oxidation. *Rare Metals, 41*(9), 3060–3068.
- <span id="page-11-3"></span>Chen, Y. H., & Lu, D. L. (2014). Amine modifcation on kaolinites to enhance CO<sub>2</sub> adsorption. *Journal of Colloid and Interface Science, 436*, 47–51.
- <span id="page-11-7"></span>Fischer, F., & Tropsch, H. (1926). The synthesis of petroleum at atmospheric pressures from gasifcation products of coal. *Brennstof-Chemie, 7*, 97–104.
- <span id="page-11-10"></span>Gao, W., & Asuha, S. (2016). Direct acid activation of kaolinite and its efects on the adsorption of methylene blue. *Applied Clay Science, 126*, 98–106.
- <span id="page-11-1"></span>Gruner, J. W. (1932). The crystal structure of kaolinite. *Zeitschrift für Kristallographie-Crystalline Materials, 83*(1–6), 75–88.
- <span id="page-11-4"></span>Hai, Y., & Asuha, S. (2015). Modifcation of acid-activated kaolinite with  $TiO<sub>2</sub>$  and its use for the removal of azo dyes. *Applied Clay Science, 114*, 558–567.
- <span id="page-11-2"></span>Hendricks, S. B. (1936). Concerning the crystal structure of kaolinite,  $Al_2O_3$ . 2SiO<sub>2</sub>. 2H<sub>2</sub>O, and the composition of anauxite. *Zeitschrift für Kristallographie-Crystalline Materials, 95*(1–6), 247–252.
- <span id="page-11-16"></span>Horváth, E., & Cseh, T. (2003). Thermal treatment of mechanochemically activated kaolinite. *Thermochimica Acta, 404*(1–2), 227–234.
- <span id="page-11-12"></span>Hu, P., & Yang, H. (2013). Insight into the physicochemical aspects of kaolins with diferent morphologies. *Applied Clay Science, 74*, 58–65.
- <span id="page-11-19"></span>Irandoust, A., & Haghtalab, A. (2017). A hybrid reduction– impregnation method in preparation of  $Co-Ru/\gamma-Al_2O_3$ catalyst for Fischer-Tropsch synthesis. *Catalysis Letters, 147*, 2967–2981.
- <span id="page-11-14"></span>Kelleher, B., & O'Dwyer, T. (2002a). Intercalation of benzamide into expanded kaolinite under ambient environmental conditions. *Clays and Clay Minerals, 50*(3), 331–335.
- <span id="page-11-15"></span>Kelleher, B., & O'Dwyer, T. (2002b). The effect of kaolinite intercalation on the structural arrangements of N-methylformamide and 1-methyl-2-pyrrolidone. *Journal of Colloid and Interface Science, 255*(2), 219–224.
- <span id="page-11-13"></span>Komori, Y., & Kuroda, K. (1998). A kaolinite-NMF-methanol intercalation compound as a versatile intermediate for further intercalation reaction of kaolinite. *Journal of Materials Research, 13*(4), 930–934.
- <span id="page-11-11"></span>Letaief, S., & Detellier, C. (2011). Single kaolinite nanometer layers prepared by an in situ polymerization–exfoliation process in the presence of ionic liquids. *Langmuir, 27*(24), 15248–15254.
- <span id="page-11-5"></span>Li, C., & Dionysiou, D. D. (2018). Acetic acid functionalized  $TiO<sub>2</sub>/k$ aolinite composite photocatalysts with enhanced photocatalytic performance through regulating interfacial charge transfer. *Journal of Catalysis, 367*, 126–138.
- <span id="page-11-18"></span>Li, X., & Komarneni, S. (2017). Methoxy-grafted kaolinite preparation by intercalation of methanol: Mechanism of its structural variability. *Applied Clay Science, 137*, 241–248.
- <span id="page-11-6"></span>Li, C., & Sun, Z. (2020). Tuning and controlling photocatalytic performance of TiO<sub>2</sub>/kaolinite composite towards ciprofoxacin: Role of 0D/2D structural assembly. *Advanced Powder Technology, 31*(3), 1241–1252.
- <span id="page-12-5"></span>Lim, Y. D., & Aditya, S. (2018). Enhanced feld emission properties of carbon nanotube bundles confined in  $SiO<sub>2</sub>$  pits. *Nanotechnology, 29*(7), 075205.
- <span id="page-12-4"></span>Orzechowski, K., & Głowinski, J. (2006). Dielectric properties of intercalated kaolinite. *Journal of Physics and Chemistry of Solids, 67*(5–6), 915–919.
- <span id="page-12-9"></span>Panda, A. K., & Singh, R. K. (2010). Effect of sulphuric acid treatment on the physico-chemical characteristics of kaolin clay. *Colloids and Surfaces A: Physicochemical and Engineering Aspects, 363*(1–3), 98–104.
- <span id="page-12-6"></span>Prieto, G., & De Jongh, P. E. (2014). Quantitative relationship between support porosity and the stability of pore-confined metal nanoparticles studied on CuZnO/SiO<sub>2</sub> methanol synthesis catalysts. *ACS Nano, 8*(3), 2522–2531.
- <span id="page-12-0"></span>Qu, H., & Yang, H. (2023). Sandwich-structured nickel/kaolinite catalyst with boosted stability for dry reforming of methane with carbon dioxide. *Chemical Engineering Journal*, *453*, 139694.
- <span id="page-12-10"></span>Scherrer, P. (1918). Bestimmung der Grösse und der inneren Struktur von Kolloidteilchen mittels Röntgenstrahlen. *Nachrichten Von Der Gesellschaft Der Wissenschaften Zu Göttingen, Mathematisch-Physikalische Klasse, 1918*, 98–100.
- <span id="page-12-12"></span>Sugahara, Y., & Kato, C. (1990). Preparation of a kaolinitepolyacrylamide intercalation compound. *Clays and Clay Minerals, 38*, 137–143.
- <span id="page-12-2"></span>Tian, L., & Ibrahim, K. E. (2020). Insight into the loading and release properties of an exfoliated kaolinite/cellulose fber (EXK/CF) composite as a carrier for oxaliplatin drug: Cytotoxicity and release kinetics. *ACS Omega, 5*(30), 19165–19173.
- <span id="page-12-14"></span>Tunney, J. J., & Detellier, C. (1997). Interlamellar amino functionalization of kaolinite. *Canadian Journal of Chemistry, 75*(11), 1766–1772.
- <span id="page-12-1"></span>Wada, K. (1961). Lattice expansion of kaolin minerals by treatment with potassium acetate. *American Mineralogist: Journal of Earth and Planetary Materials, 46*(1–2), 78–91.
- <span id="page-12-15"></span>Wang, C., & Hu, Y. H. (2019). Highly selective production of  $C_5-C_{12}$  hydrocarbons over efficient Ru/heteropoly-acid catalysts. *Fuel, 244*, 395–402.
- <span id="page-12-7"></span>Wang, J., & Zhang, P. (2017). Layered birnessite-type MnO<sub>2</sub> with surface pits for enhanced catalytic formaldehyde oxidation activity. *Journal of Materials Chemistry A, 5*(12), 5719–5725.
- <span id="page-12-11"></span>Wiewióra, A. (1969). Potassium acetate intercalation in kaolinite and its removal; efect of material characteristics. *Proceedings of the International Clay Conference, 1*, 723–733.
- <span id="page-12-8"></span>Zhang, Q., & Liu, Q. (2019). Hierarchical structure kaolinite nanospheres with remarkably enhanced adsorption properties for methylene blue. *Nanoscale Research Letters, 14*(1), 1–9.
- <span id="page-12-13"></span>Zhu, X., & Yan, C. (2016). Defects in structure as the sources of the surface charges of kaolinite. *Applied Clay Science, 124*, 127–136.
- <span id="page-12-3"></span>Zuo, X., & Yang, H. (2018). Intercalation and exfoliation of kaolinite with sodium dodecyl sulfate. *Minerals, 8*(3), 112.

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